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Lewis Acid Strength of Interfacial Metal Sites Drives CH₃OH Selectivity and Formation Rates on Cu-based CO₂ Hydrogenation Catalysts

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Abstract: CH₃OH formation rates in CO₂ hydrogenation on Cu-based catalysts sensitively depend on the nature of the support and the presence of promoters. In this context, Cu nanoparticles supported on tailored supports decorated with highly dispersed M sites on SiO₂ (Cu/M@SiO₂; M = Ti, Zr, Hf, Nb, Ta) were prepared via Surface Organometallic Chemistry, and their catalytic performance (reactivity/selectivity) was systematically investigated for the hydrogenation of CO₂ to CH₃OH. The presence of Lewis acid sites enhances CH₃OH formation rates, while CO formation rates are unaffected. Measures of the Lewis acid strength and detection of methoxy reaction intermediates indicate that the promotion of CH₃OH formation rates and selectivities likely originate from the stabilization of formate and methoxy surface intermediates at the periphery of Cu nanoparticles. The stabilization of surface intermediates depends sensitively on the Lewis acid strength of M sites, determined by measuring enthalpies of pyridine adsorption and the ¹³C chemical shifts of OCH₃ coordinated to M, which are demonstrated here to be a molecular descriptor for Lewis acidity and reactivity in CO₂ hydrogenation.

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Section S1: Experimental

General procedures

All manipulations for support preparation, organometallic synthesis, and organometallic grafting were performed under argon atmosphere in an MBRAUN UNIIab glove box or using standard Schlenk techniques. Toluene and pentane were dried by using a two-column solvent purification system (MBRAUN MB-SPS-800, equipped with MBRAUN SPS alumina columns), then stored over 4Å molecular sieve (activated by treatment under vacuum ($\sim 10^{-3}$ Pa) at 573 K for 12 h); tetrahydrofuran and C₆D₆ were distilled from Na⁰/benzophenone under argon and under vacuum ($\sim 10^{-3}$ Pa), respectively, then stored over activated 4Å molecular sieve. Celite® was placed in a Schlenk frit, dried at 423 K for 12 h before being flame dried under vacuum. NH₄Cl (AlfaAesar, 99.999 %) was dried under vacuum ($\sim 10^{-1}$ Pa) at 323 K overnight before being transferred in a glovebox. Reagents were used as received unless otherwise specified. [Cu(O'Bu)]₄,^[1] Ti(OSi(O'Bu)₃)₃(O'Pr),^[2] Hf(OSi(O'Bu)₃)₄,^[3] Zr(OSi(O'Bu)₃)₄,^[3] and Nb(OSi(O'Bu)₃)₃(O'Pr)₂,^[4] were synthesized according to literature procedures. The syntheses of [Nb(OCH₃)₅]₂,^[5] [Ta(OCH₃)₅]₂,^[6] and Ta(OSi(O'Bu)₃)₃(O'Pr)₂,^[7] were adapted from literature procedures and are reported below; the products were verified via single crystal X-ray diffraction.

Solution 1H and ^{13}C nuclear magnetic resonance (NMR) spectra (C_6D_6 , 298 K) were recorded using a Bruker 300 MHz NMR spectrometer. 1H NMR spectra analyzed for quantification used ferrocene as an internal standard and a relaxation delay (d1) of 60 s. 1H chemical shifts were referenced to the residual solvent peak at 7.16 ppm. Solution 1H and ^{13}C NMR spectra of the molecular compounds [Nb(OCH₃)₅]₂ and [Ta(OCH₃)₅]₂ were recorded on a Bruker 400 MHz AVANCE III spectrometer. The solution 1H - ^{13}C HSQC spectra were also recorded. The chemical shifts were referenced to the residual peak at 7.16 ppm (1H) and 128.06 ppm (^{13}C). NMR spectra for [Nb(OCH₃)₅]₂ and [Ta(OCH₃)₅]₂ are included in Section S10.

Infrared (IR) spectra of the molecular compounds [Nb(OCH₃)₅]₂ and [Ta(OCH₃)₅]₂ were collected under inert atmosphere using a Bruker Alpha spectrometer in attenuated total reflectance (ATR) mode, equipped with a diamond ATR module (uncooled deuterated triglycine sulfate (DTGS) detector, 2 cm⁻¹ spectral resolution, 400-4000 cm⁻¹, average of 32 scans; spectra included in Section S2). IR spectra of the catalyst solids were collected at each step of the synthesis under argon atmosphere. IR spectra of self-supporting wafers (0.01-0.02 g.cm⁻²) were recorded in transmission mode using a Bruker Alpha FT-IR spectrometer (uncooled DTGS detector, 4 cm⁻¹ spectral resolution, 400-4000 cm⁻¹, average of 24 scans; transmission module), unless stated otherwise. Spectra are normalized using the Si-O-Si overtones of the SiO₂ support for all SiO₂-containing materials.

Preparation of [Nb(OCH₃)₅]₂

To a red solution of 1.5 g of NbCl $_5$ (5.55 mmol, 1 equiv.) and 1.48 g of NH $_4$ Cl (27.7 mmol, 5 equiv.) in toluene (20 ml) cooled to 273 K, was added dropwise 20 ml of dry CH $_3$ OH. The solution immediately became colorless after addition of dry CH $_3$ OH and was stirred for 2 h at 273 K. A toluene suspension of 1.08 g of NaNH $_2$ (27.7 mmol, 5 equiv.) was added dropwise to the cold mixture then stirred for 30 min at 273 K; the mixture was warmed to 298 K and stirred for an additional hour. The colorless solution, containing a white precipitate, was filtered over Celite $_0$, and the solvent was removed in vacuo to yield a yellow oily residue. The residue was sublimed at 323 K (10^{-3} Pa). The desired material, [Nb(OCH $_3$) $_5$] $_2$, was recovered as a white solid (38% yield).

Spectra for this compound are included in the ESI, Section S10.

 1 H NMR (C₆D₆, 300 MHz): δ 4.06 ppm (6 H, μ_2 -OCH₃), 4.24 ppm (24 H, μ_1 -OCH₃); 13 C NMR (C₆D₆, 300 MHz): δ 60.5 ppm (μ_2 -OCH₃), 61.1 ppm (μ_1 -OCH₃); IR (Diamond ATR): 2915, 2888, 2816, 2801, 1451, 1430, 1150, 1061, 1018, 523, 471 cm⁻¹

Preparation of [Ta(OCH₃)₅]₂

To a yellow solution of 1.97 g of $TaCl_5$ (5.50 mmol, 1 equiv.) and 1.48 g of NH_4Cl (27.7 mmol, 5 equiv.) in toluene (20 ml) cooled to 273 K, was added dropwise 20 ml of dry CH_3OH . The solution immediately turned colorless after the addition of CH_3OH and was stirred for 1 h at 273 K. A toluene suspension of 1.08 g of $NaNH_2$ (27.7 mmol, 5 equiv.) was added dropwise to the cold mixture, which was then stirred for 30 min at 273 K. The solution was allowed to slowly warm to 298 K then stirred for 12 h at 298 K. The colorless solution, containing a white precipitate, was filtered over $Celite^{®}$. The solvent

was removed in vacuo, leaving an off-white residue. The residue was sublimed (10⁻³ Pa) at 313 K. The desired material was recovered as a white solid (42% yield).

Spectra for this compound are included in the ESI, Section S10.

 1 H NMR (C₆D₆, 300 MHz): δ 4.11 ppm (6 H, μ_2 -OCH₃), 4.34 ppm (24 H, μ_1 -OCH₃); 13 C NMR (C₆D₆, 300 MHz): δ 59.6 ppm (-OCH₃); IR (Diamond ATR): 2922, 2897, 2822, 2807, 1455, 1436, 1157, 1076, 1009, 566, 472 cm⁻¹

Preparation of Ta(OSi(O'Bu)₃)₃(O'Pr)₂

To a colorless solution of 0.5 g (0.74 mmol, 1 equiv.) of [Ta(OCH₃)₅]₂ in pentane (20 ml) was added 20 ml of dry C₃H₇OH. After stirring for 30 min. at 298 K, 1 g (3.7 mmol, 5 equiv.) Si(O'Bu)₃OH in pentane (20 mL) was added dropwise. The solution was stirred for 1 h at 298 K. The solvent was removed in vacuo. The unreacted Si(O'Bu)₃OH was removed by sublimation at 333 K (10⁻³ Pa). The product was redissolved in pentane. Concentration and cooling (238 K) of the solution afforded colorless crystals. The product was confirmed by H-NMR; the spectrum was consistent with previous reports.^[7]

Catalyst synthesis

Fumed SiO₂ (Evonik AEROSIL 200) was formed into aggregates by combining with deionized distilled water (MilliQ; resistivity > 18.2 M Ω .cm) to form a thick slurry, drying at 393 K for >24 h, then crushing and sieving to retain aggregates 250-450 μ m in size. The aggregates were treated at 973 K (0.0167 K.s⁻¹) for 12 h under flowing synthetic air (PanGas; 1.67 cm³.s⁻¹) then cooled to room temperature under high vacuum (~10⁻³ Pa) to yield SiO₂₋₇₀₀. SiO₂₋₅₀₀ was prepared from compacted SiO₂ by treatment at 773 K (0.0167 K.s⁻¹) for 12 h under flow of synthetic air (1.67 cm³.s⁻¹) followed by treatment under high vacuum (~10⁻³ Pa) while cooling to room temperature. The number of hydroxyl groups accessible for grafting for SiO₂₋₇₀₀ and SiO₂₋₅₀₀ was determined by reaction of [Mg(CH₂Ph)₂(THF)₂]^[8] with the oxide materials (in C₆D₆) and quantification of the released toluene using ¹H NMR; these values are 0.49 and 0.38 mmol OH.g⁻¹ for SiO₂₋₅₀₀ and SiO₂₋₇₀₀, respectively (corresponding to areal densities of 1.5 and 1.1 OH.nm⁻² SiO₂, respectively).

M-decorated SiO₂ supports (M@SiO₂, where M = Ti, Zr, Hf, or Nb) were prepared using the method as described next. A solution of the appropriate precursor (Ti(OSi(O'Bu)₃)₃(O'Pr), Hf(OSi(O'Bu)₃)₄, Zr(OSi(O'Bu)₃)₄, Nb(OSi(O'Bu)₃)₃(O'Pr)₂), or Ta(OSi(O'Bu)₃)₃(O'Pr)₂) in toluene (0.15 mmol; 20 mL) was added to a slurry of SiO₂₋₇₀₀ (1 g) in toluene (10 mL) then stirred for 12 h. The supernatant was decanted from the settled slurry. The material was rinsed three times with toluene then dried under vacuum (~10⁻³ Pa) at room temperature, followed by treatment under vacuum (~10⁻³ Pa) at 773 K (0.083 K.s⁻¹) for 5 h.

Cu nanoparticles were dispersed on $M@SiO_2$ and SiO_{2-500} supports: a solution of $[Cu(O^fBu)]_4$ in pentane (0.2 mmol; 20 mL) was added to a slurry of the support (1 g for $M@SiO_2$ or SiO_{2-500}) in pentane (10 mL) and stirred for 12 h. The solvent was decanted, and the solids were washed thoroughly with pentane three times. The solids were dried at 298 K (~10⁻³ Pa), treated under flowing H_2 (101 kPa, PanGas 99.999%) at 773 K (0.083 K.s⁻¹) for 5 h, then cooled to 298 K under flow of H_2 .

Catalyst characterization

Transmission electron microscopy (TEM) images of samples exposed to ambient conditions were obtained using a Hitachi HT7700 microscope (100 kV; bright field detector). Samples were prepared for imaging by depositing samples as dry fine powders onto ultrathin lacey carbon on a 300 mesh Au grids (Ted Pella) then agitating to remove the excess powder. Micrographs were processed using ImageJ to obtain the particle size distribution by assessing >150 particles. The particle size distribution was fit to lognormal distributions.

The number of surface Cu atoms (Cu_s) was determined by titration using N₂O pulses (assuming a 2:1 Cu:N₂O stoichiometry; 363 K). The sample (30-50 mg) was placed in a quartz u-tube reactor (8 mm ID) on a plug of quartz wool, with a K-type thermocouple aligned axially with the center of the catalyst bed. The reactor was placed within the BELCAT-B apparatus (BEL Japan, Inc.) and treated under a flow of 50% H₂/He (PanGas 99.999%; 0.5 cm³.s⁻¹) at 573 K for 2 h (14 K.min⁻¹) before cooling to analysis temperature. Pulses of N₂O (1% N₂O in He, PanGas; 2.6-2.8 μ LSTP N₂O per pulse) were introduced by an injection valve every 120 s. The N₂O and N₂ contents of the reactor effluent were monitored by thermal conductivity detection.

Cu, Ti, Zr, Hf, and Nb contents were determined by inductively coupled plasma – atomic emission spectroscopy (Mikroanalytisches Labor Pascher, Remagen, Germany).

IR spectra (Nicolet NEXUS 6700 infrared spectrometer; 2 cm⁻¹ spectral resolution, 1200-4000 cm⁻¹, average of 64 scans) of catalysts exposed to CO (298 K) were collected in transmission mode using an uncooled deuterated tryglycine sulfate (DTGS) KBr detector. A wafer of catalyst material (0.01-0.02 g.cm⁻¹) was pressed into an aluminum ring, which was placed into a glass vacuum cell with CaF₂ windows. The cell was exposed to 3.5 Pa CO at 298 K for 5 min; an IR spectrum was collected with the material under CO atmosphere. The cell was then evacuated (~10⁻³ Pa) for 15 minutes at 298 K, then an IR spectrum was collected.

The presence and strength of Lewis acid sites in $M@SiO_2$ were characterized by the IR spectra of adsorbed pyridine using a custom-built transmission $cell^{[9]}$ coupled to a Fourier transform-infrared (FTIR) spectrometer (Bruker, Vertex 70) with a liquid N_2 -cooled HgCdTe detector. Catalysts were pressed into self-supporting disks (~40 mg) and placed within the transmission cell, which was assembled using CaF_2 windows and connected to a gas manifold with lines heated to 423 K. All materials were first heated to 573 K at 10 K min⁻¹ and held for 2 h under flowing He (70 cm³ min⁻¹; Airgas, Ultra-zero grade), with the intent to desorb water and residual organics. Pyridine (0.1 kPa; Sigma-Aldrich, 99%) was introduced via a syringe pump (KD Scientific, Legato 100) and vaporized in the heated gas-transfer lines into a stream of flowing He (70 cm³ min⁻¹). Steady-state IR spectra (128 scans, 1 cm⁻¹ resolution) were obtained as a function of temperature (453 – 523 K) while flowing the pyridine/He stream over the samples.

Ex situ nuclear magnetic resonance (NMR) experiments were performed to examine the surface intermediates present on the catalyst materials. Samples (100 mg) were contacted with 5 bar of ¹³CO₂ and H₂ (in 1:3 ratio) in a thick-walled glass reactor for 12 h at 503 K and subsequently cooled rapidly to 77 K. The material was treated under vacuum at 298 K (10⁻³ Pa) for 0.5 h. The treated material was loaded into a 4 mm rotor for solid-state NMR in an argon atmosphere glove box without exposure to atmosphere. Solid-state NMR spectra (¹H and ¹³C) were recorded on a Bruker 400 MHz AVANCE III HD spectrometer equipped with a 4 mm magic-angle-spinning (MAS) triple resonance probe operating in double resonance mode with a MAS frequency of 10 kHz. Ramped cross-polarization (¹H-¹³C) with ¹H excitation frequency of 100 kHz was used for most experiments. The contact time was maintained at 2 ms. Adamantane was used as an external secondary reference to calibrate chemical shifts.

X-Ray absorption spectroscopy

X-Ray absorption spectra at the Nb K-edge, Hf L-edges, or Ta L-edges were measured at the SuperXAS beamline at the Swiss Light Source (SLS). The SLS operates in top-up mode at a 2.4 GeV electron energy and a current of 400 mA. The incident photon beam provided by a 2.9 T super bend magnet source was selected by a Si(111) quick-EXAFS monochromator. [10] Focusing and the rejection of higher harmonics were achieved using a rhodium-coated collimating mirror at 2.5 mrad. During the measurements, the monochromator rotated with 10 Hz frequency, and X-ray absorption spectra were collected in transmission mode using ionization chambers developed for quick data collection with 1 MHz frequency. [10] The resulting spectra were averaged over 5 minutes using ProXAS software. [111] Calibration of the monochromator energy position was performed by setting the inflection point of a Pt foil spectrum recorded simultaneous with the samples (set to 11564 eV) for Hf and Ta. The calibration for the Nb K-edge was performed using a Nb foil, setting the inflection point at 18956 eV.

Ex situ samples were pressed in pellets of optimal thickness for transmission detection. Pellets were prepared in an Ar-filled glovebox and sealed within two aluminized plastic bags (polyaniline (14 μ m), polyethylene (15 μ m), Al (12 μ m), polyethylene (75 μ m); Gruber-Folien GmbH & Co. KG) using an impulse heat sealer to avoid exposure to air. One bag was removed immediately before measurement.

EXAFS data were analyzed using Demeter software package (version 0.9.26). The data were fitted in R-space, after a Fourier Transform, using a k-weight of 3 and a window based on a Hanning function with dR = 0.5 and dk = 1. The specific ranges and values are specified below every spectrum. In the case of highly intense glitches, the k-range was adjusted to exclude them from Fourier Transform; glitches at k < 13 were removed manually. The respective scattering paths were generated based on available crystallographic data (Hf) or DFT-derived geometry-optimized models (Ta, Nb). The parameters R, σ , and Δ E were optimized during the fitting procedure, while the parameter N was fixed. The value of S_0^2 was obtained from fitting the molecular compounds. The amount of fitted parameters was below the 2/3 of the Nyquist criterion.

CO₂ hydrogenation reactions

Rates of CO_2 hydrogenation were measured at 503 K at differential reactant conversions (< 7%) on all samples (0.20-0.25 g), diluted with 5 g SiC (Fisher Scientific) and held against a sintered metal frit, axially centered within a tubular reactor (316 S.S., 9.1 mm i.d.) with plug-flow hydrodynamics. Temperatures were measured with a K-type thermocouple held at the midpoint of the packed bed and maintained using a resistively heated furnace with electronic controllers. Samples were treated at 573 K (0.083 K.s⁻¹) in H₂ flow (1 cm³.s⁻¹; PanGas 99.999% H₂) for 1 h before cooling to reaction temperatures. CO_2 , H_2 , and Ar (CarbaGas; mixed in a 1:3:1 CO_2 :H₂:Ar molar ratio) were introduced using electronic mass flow controllers (Bronkhorst EL-FLOW); total flow rates were varied between 6 and 100 cm³.min⁻¹ to vary reactant residence time without changes in individual reactant pressures. Transfer lines were heated (>393 K) to prevent product condensation. Reactor pressure was maintained at 2.6 MPa using a back-pressure regulator (Bronkhorst EL-PRESS). Reactant and product concentrations in the reactor effluent were measured using an on-line gas chromatograph (Agilent 7890N GC) equipped with Restek Rt-U-BOND (30 m x 0.53 mm x 20 μ m) and Rt-Msieve 5A (30 m x 0.53 mm x 50 μ m) columns, flame ionization detection (for CH₃OH and dimethylether), and thermal conductivity detection (for CO_2 , H_2 , Ar, and CO). Products measured were primarily CO and CH₃OH. Trace amounts of dimethylether (<0.005 selectivity), the bimolecular dehydration product of CH₃OH, were detected for Cu/Ti@SiO₂ and are accounted for in CH₃OH rates and selectivities (assuming 2:1 CH₃OH:dimethylether). Methane concentrations were below detection limits at all conditions on all samples.

Reaction rates for SiO₂ and M@SiO₂ materials were negligible (all product concentrations were below detection limits). Rates and selectivities for physically mixed Cu/SiO₂ and M@SiO₂ (1:1 mass ratio) were the same as for Cu/SiO₂ alone. Rates are calculated on a per-carbon basis and are normalized by the mass of Cu in each catalyst. Product selectivity is defined on a per-C basis. Negligible catalyst deactivation (<5% decrease in rates after 48 h) was detected by periodically measuring rates at a reference condition.

CH₃OH formation rates decreased as a function of increasing residence time for all samples except Cu/SiO₂. The rigorous extrapolation of rates to zero residence times to obtain initial rates requires knowledge of the functional form of the rate expression; in its absence, rates at the shortest residence times, where product inhibition is the least severe because of the low partial pressures of products at small reactant conversions, were extrapolated exponentially to obtain initial values. CO formation rates are nearly independent of residence time; their averaged values were taken as initial CO formation rates.

Section S2: Infrared spectra during catalyst synthesis

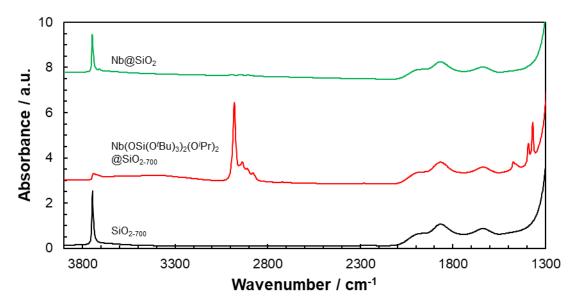


Figure S1a. Infrared spectra of solids during synthesis of Nb single sites on SiO₂ (Nb@SiO₂). SiO₂₋₇₀₀ (bottom), after grafting of Nb(OSi(O'Bu)₃)₃(O'Pr)₂ on SiO₂ and rinsing and drying (middle), and after thermal treatment under vacuum (top).

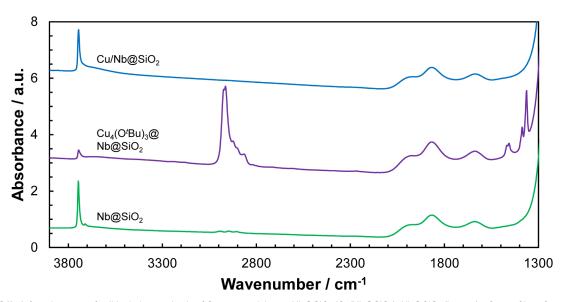


Figure S1b. Infrared spectra of solids during synthesis of Cu nanoparticles on $Nb@SiO_2$ (Cu/Nb@SiO₂). $Nb@SiO_2$ (bottom), after grafting of $[Cu(O'Bu)]_4$ on $Nb@SiO_2$ and rinsing and drying (middle), and after thermal treatment under H_2 flow (top).

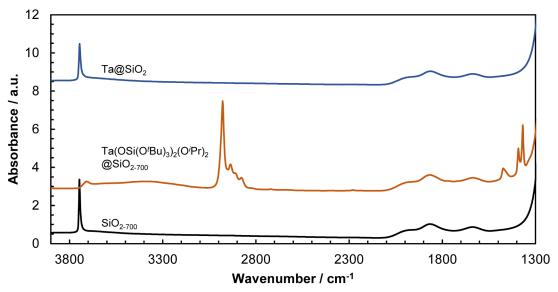
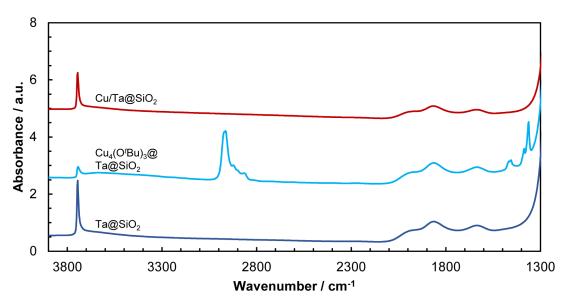


Figure S2a. Infrared spectra of solids during synthesis of Ta single sites on SiO₂ (Ta@SiO₂). SiO₂₋₇₀₀ (bottom), after grafting of Ta(OSi(O'Bu)₃)₃(O'Pr)₂ on SiO₂ and rinsing and drying (middle), and after thermal treatment under vacuum (top).



 $\textbf{Figure S2b.} \ \, \text{Infrared spectra of solids during synthesis of Cu nanoparticles on } \ \, \text{Ta@SiO}_2 \, (\text{Cu/Ta@SiO}_2). \ \, \text{Ta@SiO}_2 \, (\text{bottom}), \ \, \text{after grafting of } \ \, \text{[Cu(O'Bu)]}_4 \, \text{on } \ \, \text{Ta@SiO}_2 \, \text{and rinsing and drying (middle), and after thermal treatment under } \ \, \text{H}_2 \, \text{flow (top)}.$

Section S3: Particle size distributions and sample micrographs

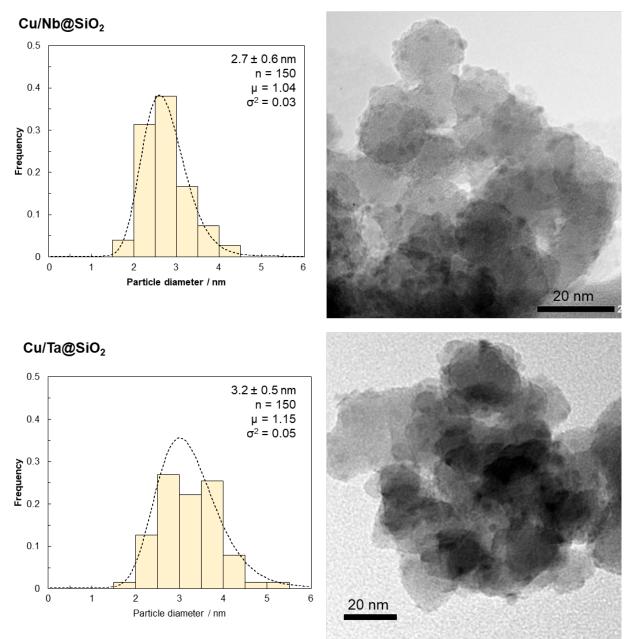


Figure S3. Cu particle size distribution and sample micrographs for Cu/Nb@SiO₂ (top) and Cu/Ta@SiO₂ (bottom). Number of particles (n) and fitting parameters (μ and σ^2) for the log-normal distributions are reported, inset in the plots.

Section S4: X-Ray absorption spectroscopy (XAS): near edge structure (XANES) and fine structure (EXAFS) spectra, fits, and analysis

In order to assess the isolation and the oxidation state of M (M = Hf, Ta, and Nb), XAS spectra were collected for assynthesized $M@SiO_2$ and $Cu/M@SiO_2$ materials (without exposure to air), as well as for $Cu/M@SiO_2$ after catalytic testing ("spent", without exposure to air). Reference spectra (molecular precursors; Ta and Nb foils) were also recorded.

Normalized X-ray absorption near edge structure (XANES) spectra at the Nb K-edge are shown in Figure S4a for Nb@SiO₂, Cu/Nb@SiO₂, and Cu/Nb@SiO₂ after reaction (spent). Spectra for the Nb molecular precursor, Nb(O'Pr)₂(OSi(O'Bu)₃)₃, and a Nb foil are also shown. The first and second derivatives of the normalized absorbance are shown in Figures S4b and SX1c, respectively. The edge energy (E₀) is a sensitive indicator of oxidation state, [1] and its position is indicated by an inflection point (denoted in Figures S4b and S4c). This value (19003 eV) is nearly identical for Nb(O'Pr)₂(OSi(O'Bu)₃)₃, Nb@SiO2, Cu/Nb@SiO2, and Cu/Nb@SiO2 (spent), indicating that the Nb oxidation state is the same among these materials (Nb(V)). In contrast, the Nb foil has a lower E₀ value (18986 eV). [2] The pre-edge features (indicated by the inflection point at 18990 eV in Figures S4b and S4c) for Nb(O'Pr)₂(OSi(O'Bu)₃)₃, Nb@SiO₂, Cu/Nb@SiO₂, and Cu/Nb@SiO₂ (spent) arise from 1s -> 4d electronic transitions and are only observable when p and d orbitals are hybridized. [1] Thus, increased centrosymmetry of the ligand field at the metal center (e.g., for Oh, compared to Td) leads to a decrease in the intensity of this pre-edge feature. The intensity of the pre-edge increases from the molecular precursor to Nb@SiO2 and further to Cu/Nb@SiO₂, likely reflecting the undercoordinated state of Nb in these materials (relative to the molecular precursor; the structure of Nb(O[']Pr)₂(OSi(O[']Bu)₃)₃ from force-field geometry optimizations has a nearly O_h geometry, similar to the crystal structure reported for Ta(OEt)₂(OSi(O'Bu)₃)₅^[7]). The intensity of the pre-edge decreases significantly from the fresh Cu/Nb@SiO2 material to the spent catalyst, reflecting the adsorption of surface species (methoxy, formate, H2O) to these metal centers, concomitantly increasing the symmetry of their ligand fields.

To examine the well-dispersed nature of Nb metal centers, the extended X-ray absorption fine-structure (EXAFS) spectra were analyzed (Fig. S4d). Nb K-edge EXAFS spectra are shown in Figure S4d and were fit to obtain coordination numbers and the first and second coordination sphere. Tables S1-S4 show the fitting parameters for the Nb-containing materials. In no cases were Nb-Nb or Nb-O-Nb scattering paths able to be fit, with attempts to fit these spectra with Nb-Nb or Nb-O-Nb paths leading to extremely low degeneracy or high Debye-Waller factors, indicating that the presence of aggregated Nb is unlikely. Furthermore, the very low intensity of distant shells for all Nb@SiO₂ materials indicates that these cannot be attributed to Nb-Nb or Nb-O-Nb scattering paths. Together, these data indicate that Nb metal centers remain highly dispersed in Nb@SiO₂, Cu/Nb@SiO₂, and Cu/Nb@SiO₂ (spent). Additionally, Nb-Cu (and related) paths could not be fit for both Cu/Nb@SiO₂ and Cu/Nb@SiO₂ (spent). It is feasible that distant shells could arise from Cu scattering from proximal Cu nanoparticles, which is challenging to ascertain because of the absence of appropriate reference spectra. Nevertheless, these EXAFS fits, together with the overall similarity of XANES spectra among Nb@SiO₂, Cu/Nb@SiO₂, and Cu/Nb@SiO₂ (spent), indicate that Cu-Nb alloys are not formed and that for all materials, Nb is well-dispersed and retains a +5 oxidation state.

Similar trends in oxidation state (L₁-edge XANES) and site isolation (L₃-edge EXAFS) are observed for Ta and Hf. The normalized XANES spectra at the Ta L₁-edge is shown in Figure S5a for $Ta(O'Pr)_2(OSi(O'Bu)_3)_3$, $Ta@SiO_2$, $Cu/Ta@SiO_2$, and $Cu/Ta@SiO_2$ (spent), as well as for a Ta foil reference. The first and second derivatives of the normalized absorbance for the same materials are shown in Figures S5b and S5c, respectively. The edge energies for $Ta(O'Pr)_2(OSi(O'Bu)_3)_3$, $Ta@SiO_2$, $Cu/Ta@SiO_2$, and $Cu/Ta@SiO_2$ (spent) are nearly identical (11699 eV) and greater than that for the Ta foil (11684 eV), indicating that for all of these materials, Ta is in the same oxidation state (Ta(V)). For 5d metals, the pre-edge of the L₁-edge reflects 2s -> 5d electronic transitions and thus can be interpreted analogously to the pre-edge of the K-edge for 4d metals^[1] (vide supra for Nb K-edge). The intensity of the pre-edge feature, indicated by the inflection point at 11686 eV in Figures S5b and S5c, increases from $Ta(O'Pr)_2(OSi(O'Bu)_3)_3$ to $Ta@SiO_2$ and to $Cu/Ta@SiO_2$, before decreasing in intensity for $Cu/Ta@SiO_2$ (spent). These trends are the same as those observed for the Nb pre-edge and indicate the decrease in the centrosymmetry of the Ta metal center from the molecular precursor (which is nearly O_h in geometry, according to force-field geometry optimizations of a structure derived from the $Ta(OEt)_2(OSi(O'Bu)_3)_3$ crystal structure^[7]) to the isolated metal center on SiO_2 (in $Ta@SiO_2$ and in $Cu/Ta@SiO_2$). The pre-edge intensity decreases for the spent catalyst because of the adsorption of surface methoxy or formate species (or H_2O) onto Ta metal centers, increasing the symmetry of their ligand fields.

EXAFS fits and parameters for the Ta-containing materials (Ta L₃-edge) are shown in Figure S5d and Tables SX5-SX8, respectively. In all cases, Ta metal centers were determined to be well-dispersed. Scattering paths for Ta-Ta or Ta-O-Ta were not able to be fit for any material. Additionally, the very low intensity of distant shells for all Ta@SiO₂ materials indicate

that these cannot be attributed to scattering from distant Ta. It is possible that these distant shells arise from scattering by Cu, which has a much smaller atomic number than Ta. Together with XANES, these data indicate that Ta metal centers remain highly dispersed Ta(V) in Ta@SiO₂, Cu/Ta@SiO₂, and Cu/Ta@SiO₂ (spent).

The normalized XANES spectra at the Hf L₁-edge is shown in Figure S6a for Hf(OSi(O¹Bu)₃)₄, Hf@SiO₂, Cu/Hf@SiO₂, and Cu/Hf@SiO₂ (spent); the first and second derivatives of the normalized absorbance for the same materials are shown in Figures S6b and S6c, respectively. The edge energy for all Hf-containing materials is nearly identical (11284 eV), and greater than the reported value for the Hf metal (11270 eV), indicating that Hf is in a +4 oxidation state for all of these materials. The intensity of the pre-edge feature (at 11273 eV) is most pronounced for Hf(OSi(O¹Bu)₃)₄, consistent with its reported crystal structure^[3] that is composed of a 1:1 mixture of Hf with four equivalent OSi(O¹Bu)₃ ligands and Hf with three equivalent ligands and one bi-dentate (κ^2) OSi(O¹Bu)₃ ligand. The intensity of the pre-edge decreases from the molecular precursor to Hf@SiO₂, Cu/Hf@SiO₂, and Cu/Hf@SiO₂ (spent), respectively. These changes indicate the increase of the centrosymmetry of the ligand field at the Hf metal center from the molecular precursor to all of the SiO₂-supported materials. The spent Cu/Hf@SiO₂ material has only a subtle pre-edge feature, reflecting the coordination of surface methoxy and formate species (or H₂O).

The EXAFS spectra at the Hf L_3 -edge are shown in Figure S6d, with fits and parameters in Tables S9-S12. Similar to the EXAFS analysis for Nb and Ta, the Hf L_3 -edge EXAFS indicates that Hf metal centers are well-dispersed. Hf-Hf and Hf-O-Hf scattering paths could not be adequately fit for any of these materials; attempts to fit these spectra with Hf-Hf and Hf-O-Hf paths led either to extremely low degeneracies or high Debye-Waller factors, indicating that the presence of aggregated Hf is unlikely. Thus, the very low intensity of distant shells for all SiO₂-supported materials renders these unlikely to arise from scattering by distant Hf metal centers but possibly from scattering by light elements (e.g., Cu). Taken together with the information from the XANES spectra, these trends indicate that Hf metal centers are highly dispersed Hf(IV) in Hf@SiO₂, Cu/Hf@SiO₂, and Cu/Hf@SiO₂ (spent).

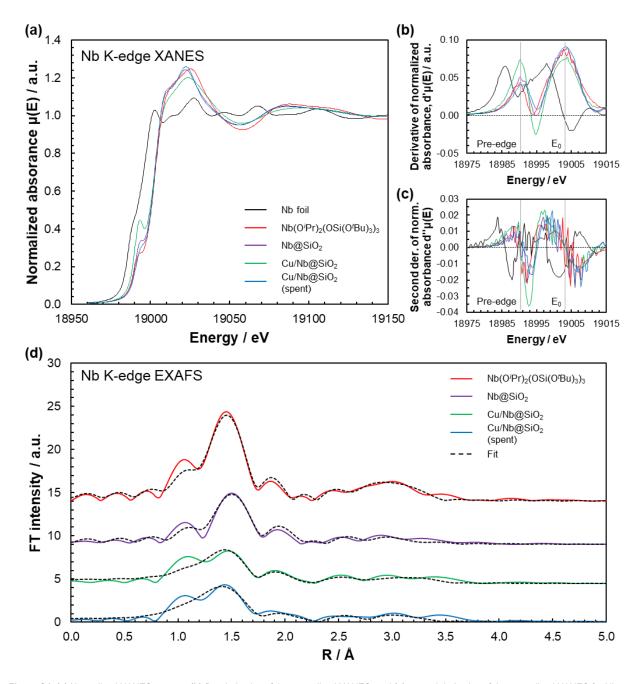


Figure S4. (a) Normalized XANES spectra, **(b)** first derivative of the normalized XANES, and **(c)** second derivative of the normalized XANES for Nb-containing materials (Nb K-edge). **(d)** EXAFS spectra (k³-weighted) and fits for Nb-containing materials (Nb K-edge). Fitting parameters and fitting windows are included in Tables S1-S4.

Table S1. Fitting parameters of the Nb K-edge EXAFS spectrum of Nb(O'Pr)2(OSi(O'Bu)3)3.

Scattering path	N	σ^2 / Å ²	R/Å
Nb-O	4	0.00294 ± 0.00046	1.910
Nb-O	1	0.00476 ± 0.00449	2.107
Nb-Si	1	0.01093 ± 0.00676	2.769
Nb-O-Si	2	0.00290 ± 0.00281	3.514
Nb-Si	2	0.00713 ± 0.00656	3.642

3 < k < 14.1, $S_0^2 = 1$, R-score = 0.01519, $\Delta E = -2.535$ eV; 1 < R < 3.6

The fitted spectrum of $Nb(O^iPr)_2(OSi(O^iBu)_3)_3$ provides a good agreement with the presumed molecular structure, which was constructed in analogy to that for $Ta(OEt)_2(OSi(O^iBu)_3)_3^{[7]}$ and subsequently geometrically optimized using force fields. This indicates that (i) amplitude reduction factor S_0^2 approaches 1 and that (ii) the molecule preserved its integrity during the XAS experiment.

Table S2. Fitting parameters of the Nb K-edge EXAFS spectrum of Nb@SiO2.

Scattering path	N	σ^2 / Å ²	R/Å
Nb-O	2.1	0.00178 ± 0.00064	1.953
Nb-O	1.9	0.00840 ± 0.00423	2.152
Nb-Si	2.1	0.01897 ± 0.00911	2.821
Nb-O-Si	3.3	0.00926 ± 0.00861	3.598

3.2 < k < 13.8, $S_0^2 = 1$, R-score = 0.0049, $\Delta E = 7.44$ eV; 1 < R < 3.1 Å

Table S3. Fitting parameters of the Nb K-edge EXAFS spectrum of Cu/Nb@SiO2.

Scattering path	N	σ^2 / \mathring{A}^2	R/Å
Nb-O	3.2	0.00759 ± 0.00135	1.949
Nb-O	2	0.00941 ± 0.00443	2.144
Nb-Si	0.9	0.00942 ± 0.00541	2.794
Nb-O-Si	2.7	0.01088 ± 0.00653	3.685

2.9 < k < 13.5; $S_0^2 = 1$, R-score = 0.06954, $\Delta E = 2.47$ eV, 1 < R < 3.2

Table S4. Fitting parameters of the Nb K-edge EXAFS spectrum of Cu/Nb@SiO2 (spent).

Scattering path	N	σ^2 / Å ²	R/Å
Nb-O	3.9	0.00985 ± 0.00095	1.919
Nb-Si	2.5	0.01658 ± 0.00474	2.782
Nb-O-Si	2.3	0.00377 ± 0.00360	3.325
Nb-O-Si	2.7	0.00563 ± 0.00181	3.503

2.8 < k < 13.8; $S_0^2 = 1$, R-score = 0.0447, $\Delta E = -9.85$ eV, 1 < R < 3.2

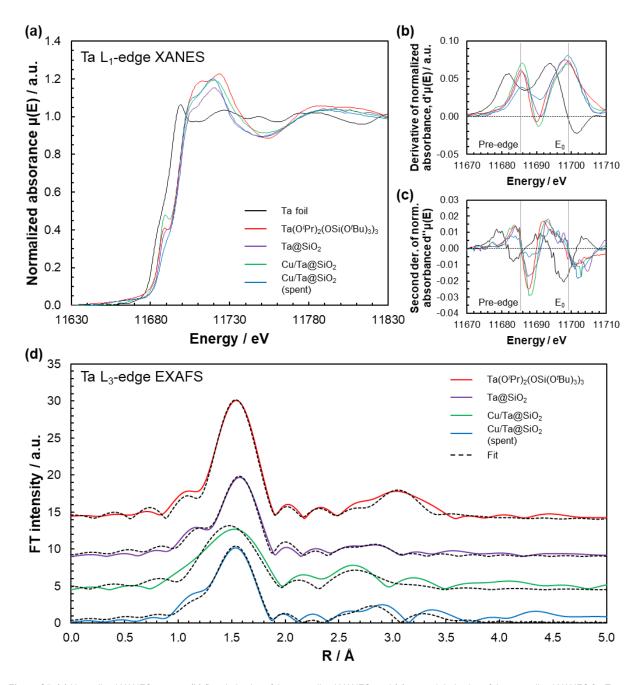


Figure S5. (a) Normalized XANES spectra, (b) first derivative of the normalized XANES, and (c) second derivative of the normalized XANES for Tacontaining materials (Ta L₁-edge). (d) EXAFS spectra (k³-weighted) and fits for Ta-containing materials (Ta L₃-edge). Fitting parameters and fitting windows are included in Tables S5-S8.

Table S5. Fitting parameters of the Ta L₃-edge EXAFS spectrum of Ta(O'Pr)₂(OSi(O'Bu)₃)₃.

Scattering path	N	σ^2 / Å ²	R/Å
Ta-O	5	0.00249 ± 0.00025	1.903
Ta-O	1	0.00476 ± 0.00340	2.296
Ta-O-Si	3.5	0.00163 ± 0.00127	3.504
Ta-Si	3	0.01252 ± 0.01241	3.729

2.8 < k < 12.8, $S_0^2 = 1$, R-score = 0.00620, $\Delta E = 8.08$ eV, 1 < R < 3.3

The fitted spectrum of $Ta(O^iPr)_2(OSi(O^iBu)_3)_3$ provides a good agreement with the presumed molecular structure, which was constructed in analogy to that for $Ta(OEt)_2(OSi(O^iBu)_3)_3^{[7]}$ and subsequently geometrically optimized using force fields. This indicates that (i) amplitude reduction factor S_0^2 approaches 1 and that (ii) the molecule preserved its integrity during the XAS experiment.

Table S6. Fitting parameters of the Ta L_3 -edge EXAFS spectrum of **Ta@SiO₂**.

Scattering path	N	σ^2 / Å ²	R/Å
Ta-O	3	0.00178 ± 0.00057	1.920
Ta-O	1.8	0.00681 ± 0.00260	1.810
Ta-Si	2	0.00780 ± 0.00268	3.212

3 < k < 13.8, $S_0^2 = 1$, R-score = 0.00999, $\Delta E = 6.722$ eV, 1 < R < 2.9

Table S7. Fitting parameters of the Ta L_3 -edge EXAFS spectrum of **Cu/Ta@SiO₂**.

Scattering path	N	σ^2 / Å ²	R/Å
Ta-O	4.5	0.00549 ± 0.00140	1.901
Ta-Si	2.6	0.00492 ± 0.00449	3.207

2.6 < k < 9.5, $S_0^2 = 1$, R-score = 0.06554, $\Delta E = 8.13$ eV, 1 < R < 3.3

n.b.: The spectrum for $Cu/Ta@SiO_2$ had a relatively low edge jump, resulting in monochromator 'glitches' of large amplitude at k > 9.5. As a result, the fitting window was constrained to a smaller width for this spectrum, and only two scattering paths were fitted. Nevertheless, the spectrum is very similar to that for $Cu/Ta@SiO_2$ (spent), which was of sufficient quality for an extended fit, suggesting the absence of Ta agglomerates.

Table S8. Fitting parameters of the Ta L_3 -edge EXAFS spectrum of Cu/Ta@SiO₂ (spent).

Scattering path	N	σ^2 / \mathring{A}^2	R/Å
Ta-O	4.6	0.00523 ± 0.00044	1.914
Ta-Si	3	0.00824 ± 0.00214	3.264
Ta-O-Si	2.1	0.00505 ± 0.00467	3.804

3 < k < 13.8, $S_0^2 = 1$, R-score = 0.02498, $\Delta E = 8.06$ eV, 1 < R < 3.4

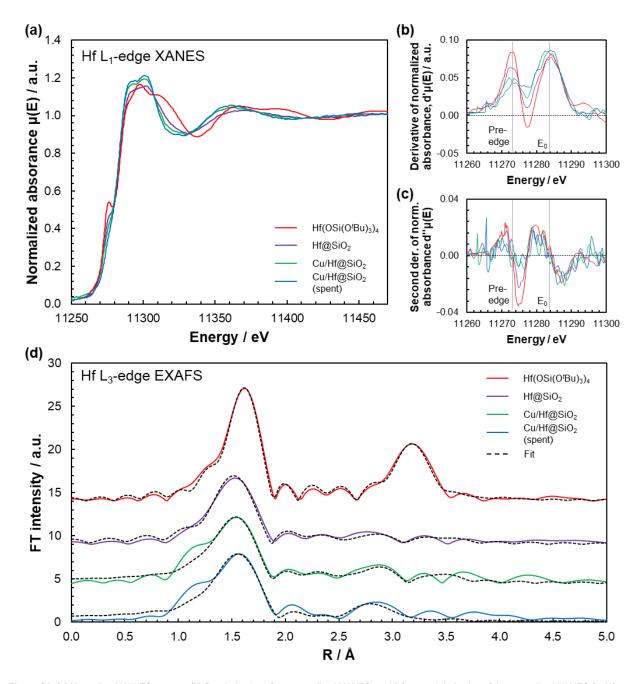


Figure S6. (a) Normalized XANES spectra, (b) first derivative of the normalized XANES, and (c) second derivative of the normalized XANES for Hf-containing materials (Hf L₁-edge). (d) EXAFS spectra (k³-weighted) and fits for Hf-containing materials (Hf L₃-edge). Fitting parameters and fitting windows are included in Tables S9-S12.

Table S9. Fitting parameters of the Hf L₃-edge EXAFS spectrum of Hf(OSi(O'Bu)₃)₄.

Scattering path	N	σ^2 / Å ²	R/Å
Hf-O	4	0.00289 ± 0.00020	1.955
Hf-O	0.5	0.00459 ± 0.00405	2.380
Hf-Si	0.5	0.00283 ± 0.00212	2.992
Hf-O-Si	3	0.00139 ± 0.00045	2.557
Hf-Si	3.5	0.01600 ± 0.00871	3.685

2.8 < k < 16, $S_0^2 = 1$, R-score = 0.00761, $\Delta E = 8.135$ eV, 1 < R < 3.3

The fitted spectrum of $Hf(OSi(O^tBu)_3)_4$ provides a good agreement with the crystallographic data, which shows a 1:1 mixture of 4- and 5-coordinate Hf (the latter with one κ^2 $OSi(O^tBu)_3$ ligand). This indicates that (i) the amplitude reduction factor So^2 approaches 1 and that (ii) the molecule preserves its integrity during the XAS experiment.

Table S10. Fitting parameters of the Hf L_3 -edge EXAFS spectrum of **Hf@SiO₂**.

Scattering path	N	σ^2 / Å ²	R/Å	
Hf-O	3.8	0.00491 ± 0.00051	1.932	
Hf-Si	2.9	0.01489 ± 0.00407	2.940	
Hf-O-Si	1.9	0.00432 ± 0.00349	3.547	
Hf-O-Si	2.5	0.00508 ± 0.00488	3.732	

2.8 < k < 12.6, $S_0^2 = 1$, R-score = 0.01751, $\Delta E = -0.342$ eV, 1 < R < 3.2

Table S11. Fitting parameters of the Hf L_3 -edge EXAFS spectrum of **Cu/Hf@SiO₂**.

Scattering path	N	σ^2 / Å ²	R/Å
Hf-O	5.4	0.00777 ± 0.00073	1.959
Hf-Si	2.2	0.00988 ± 0.00335	2.935
Hf-O-Si	2.4	0.00360 ± 0.00307	3.738
Hf-O-Si	2.0	0.00303 ± 0.00238	3.565

2.7 < k < 13.3, $S_0^2 = 1$, R-score = 0.03371, $\Delta E = -2.080$ eV, 1 < R < 3.4

Table S12. Fitting parameters of the Hf L₃-edge EXAFS spectrum of Cu/Hf@SiO₂ (spent).

Scattering path	N	σ^2 / Å ²	R/Å
Hf-O	6	0.00818 ± 0.00093	1.984
Hf-Si	2	0.01095 ± 0.00684	2.979
Hf-Si	1.6	0.00630 ± 0.00573	3.295
Hf-O-Si	3.8	0.01390 ± 0.00942	3.704

2.7 < k < 12.7, $S_0^2 = 1$, R-score = 0.03982, $\Delta E = 0.031$ eV, 1 < R < 3.3

Section S4: Infrared spectra for catalysts with adsorbed CO

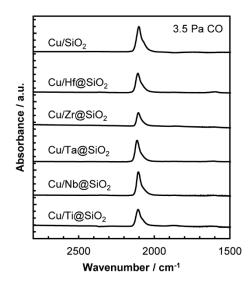
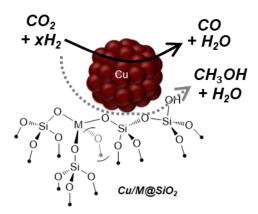


Figure S7. Difference IR spectra for Cu/SiO_2 and $Cu/M@SiO_2$ (M = Ti, Zr, Hf, Nb, or Ta) with adsorbed CO (3.5 Pa CO, 298 K), compared to samples under vacuum. Absorbance features for v(C-O) of atop-bound CO completely disappeared after vacuum treatment following adsorption of CO, which is consistent with the reversible adsorption of CO on Cu.

Section S5: Scheme of site requirements for CO₂ hydrogenation pathways



Scheme S1. Schematic representation of CO_2 hydrogenation pathways on $Cu/M@SiO_2$. CO and CH_3OH are formed in parallel reactions. CO is formed on the same site as Cu/SiO_2 (likely a Cu nanoparticle, indicated by the black arrow), while CH_3OH is formed on a Cu-M interfacial site (indicated by the grey dashed arrow).

Section S7: CH₃OH and CO formation rates for Cu/M@SiO₂ and Cu/SiO₂ catalysts

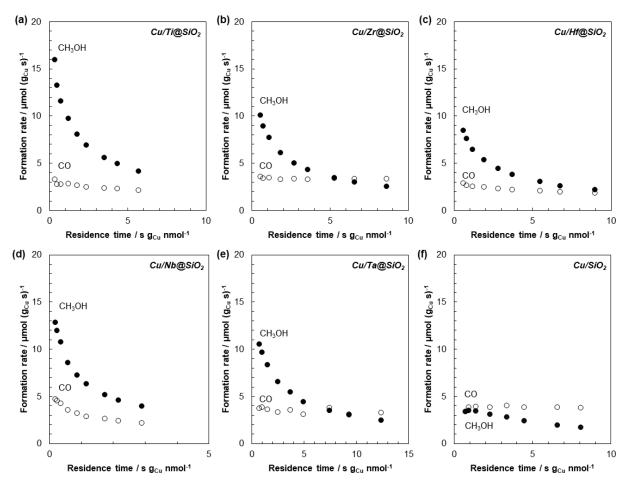


Figure S8. CH_3OH (filled) and CO (unfilled) formation rates as a function of residence time for (a) $Cu/Ti@SiO_2$, (b) $Cu/Zr@SiO_2$, (c) $Cu/Hf@SiO_2$, (d) $Cu/Nb@SiO_2$, (e) $Cu/Ta@SiO_2$, and (f) Cu/SiO_2 .

Table S13. Reaction rates on M@SiO₂ and physical mixtures of Cu/SiO₂+M@SiO₂ (1:1 mass).

Catalyst	Initial CH ₃ OH formation rate / µmol (g _{Cu} s) ^{-1 [a]}	Initial CO formation rate / µmol (gcu s)-1 [b]
Ti@SiO ₂	[c]	[c]
Nb@SiO ₂	[c]	[c]
Ta@SiO ₂	[0]	[c]
Zr@SiO ₂	[0]	[c]
Hf@SiO ₂	[0]	[c]
Cu/SiO ₂ + Ti@SiO ₂	3.65 ± 0.12	3.80 ± 0.02
Cu/SiO ₂ + Nb@SiO ₂	3.59 ± 0.17	3.67 ± 0.13
Cu/SiO ₂ + Ta@SiO ₂	3.63 ± 0.13	3.80 ± 0.05
Cu/SiO ₂ + Zr@SiO ₂	3.57 ± 0.10	3.78 ± 0.07
Cu/SiO ₂ + Hf@SiO ₂	3.54 ± 0.15	3.70 ± 0.10

[a] error represents 95% confidence interval. [b] error represents standard error. [c] below detection limits of instrument.

Section S8: Infrared spectra of catalysts with adsorbed pyridine, isobars, and pyridine adsorption enthalpies

For Cu/M@SiO₂ materials (M = Ti, Zr, Hf), two additional bands are seen compared to the spectra for their respective M@SiO₂ materials (1600 cm⁻¹, 1480 cm⁻¹), similar to those observed for Cu/SiO₂. Subtracting the pyridine adsorption band at 1444 cm⁻¹ for Cu/SiO₂ from the spectra of Cu/M@SiO₂ materials, assuming this band arises from adsorption on the Cu metal, yields areas and FWHM of the remaining bands that are nearly the same as for the respective M@SiO₂ materials (Table S14). For Ta and Nb materials, the Cu/SiO₂ adsorption bands are not observed; nevertheless, the areas and FWHM of the band at 1450 cm⁻¹ are similar despite the presence of Cu (Table S14). These data together indicate that M@SiO₂ sites are essentially identical to those after Cu nanoparticles have been dispersed on the support. Thus, the pyridine band areas measured for M@SiO₂, and therefore for Cu/M@SiO₂, are the appropriate quantity to measure.

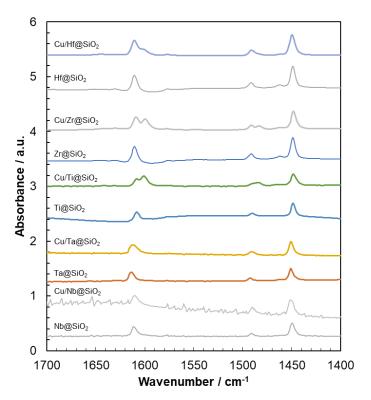


Figure \$9. Difference IR spectra for materials with pyridine adsorbed then desorbed at 373 K and spectra prior to absorption, normalized by the area of the Si-O-Si overtone bands, for M@SiO₂ and Cu/M@SiO₂ materials.

Table S14. Band area and FWHM for the pyridine vibrational band at 1450 cm⁻¹.

Band area FWHM	Cu/Ti@SiO₂ 2.1 5.7	Cu/Zr@SiO₂ 1.5 5.8	Cu/Hf@SiO₂ 3.0 5.8	Cu/Nb@SiO ₂ 3.0 5.7	Cu/Ta@SiO ₂ 3.7 5.8
Band area FWHM	Ti@SiO₂ 2.0 5.7	Zr@SiO₂ 1.6 5.8	Hf@SiO₂ 2.8 5.8	Nb@SiO₂ 2.9 5.8	Ta@SiO₂ 3.8 5.8

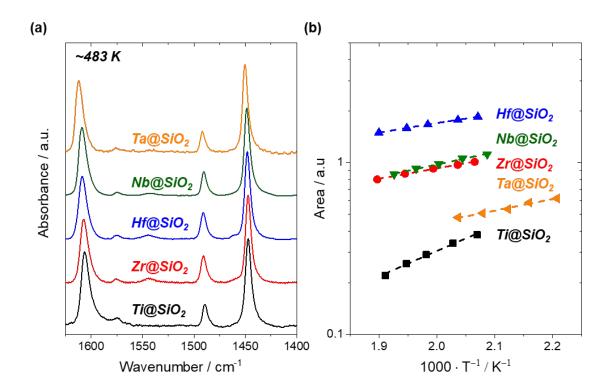


Figure \$10. (a) Normalized IR spectra of M@SiO₂ (treated by flowing He for 2 h at 573 K) with adsorbed pyridine (483 K, 0.1 kPa pyridine). (b) Isobars (0.1 kPa pyridine) for M@SiO₂ (453-523 K), using the areas of the pyridine bands at 1450 cm⁻¹. Dashed lines show the fit of the van't Hoff equation.

Table S15. Pyridine adsorption enthalpies for M@SiO₂, determined from isobars (0.1 kPa pyridine; 483-523 K). Errors represent standard error. **Material** $\Delta H_{\text{ads,pyridine}}$ / **kJ mol**-1

Material	ΔH _{ads,pyridine} / kJ m
Ti@SiO ₂	-29 ± 1
Nb@SiO ₂	-15 ± 1
Ta@SiO ₂	-13 ± 1
Zr@SiO ₂	-12 ± 1
Hf@SiO ₂	-10 ± 1

Section S9: DFT Calculation of ¹³C NMR Chemical Shifts, Comparison with Measured Values, and Natural Chemical Shielding Analysis

Computational Details

Geometry optimizations and vibrational analysis were performed with the B3LYP^[13] functional in combination with a 6-311G(d)^[14,15] basis set (for C, H, Si and O) and the Stuttgart-Dresden effective core potential (SDD)^[16,17] (for Ti, Zr, Hf, Nb, Ta) and the associated basis set using Gaussian09 (revision D.01).^[18] NMR calculations on metal complexes were performed at the same level of theory with the Gauge-Independent Atomic Orbital (GIAO)^[19,20] method and an ultrafine integration grid. The ¹³C chemical shift was referenced to tetramethylsilane (TMS; set to 0 ppm) from the calculated chemical shielding according to:

 $\delta_{iso} = \delta_{TMS} - \delta_{iso}$ (Eq. 1) where δ_{iso} is the isotropic ¹³C chemical shift, δ_{TMS} is the chemical shielding of TMS (0 ppm), and δ_{iso} is the calculated ¹³C chemical shielding.

Natural chemical shielding (NCS) analysis of the optimized cluster models was performed with ADF 2014^[21,22] with the Gauge-Independent Atomic Orbital (GIAO) method and a full electron TZP^[23] basis set and revPBE functional.^[24] Relativistic effects were included by the two component zero order regular approximation (ZORA) approach including spin-orbit coupling.^[25] The NMR chemical shielding tensors components were deconvoluted into the contributions of the individual natural localized molecular orbitals (NLMO)^[26,27] obtained by the NBO 6.0 program^[28] in its scalar-relativistic framework. Chemical shift tensors were plotted using a Mathematica script provided by Autschbach and coworkers.^[29,30]

¹³C Chemical Shift Calculations

In order to understand the measured 13 C chemical shifts of methoxy species in ssNMR, two series of methoxy species were optimized and their respective chemical shifts were calculated using DFT methods: (1) molecular models of methoxy complexes, and (2) methoxy coordinated to cluster models of SiO₂ with a metal site, coordinated by three (Ti, Zr, Hf) or four (Nb, Ta) siloxy ligands and one methoxy ligand; these SiO₂-based cluster models are used for NCS analysis in the following section. The results are summarized in Table S16. CH₃OH shows a chemical shift at δ_{iso} = 54 ppm and is close to the experimentally measured value (δ_{iso} = 49 ppm, measured in $C_6D_6^{[31]}$). The calculated chemical shift for CH₃OH is similar to that for Si(OCH₃)₄ (δ_{iso} = 53 ppm), indicating that measured signals at similar chemical shifts could be interpreted as either species. Measured chemical shifts for [Nb(OCH₃)₅]₂ (δ_{iso} = 61.1 and 60.5 ppm, measured in C_6D_6 ; Figure S26), [Ta(OCH₃)₅]₂ (δ_{iso} = 59.6 ppm, measured in C_6D_6 ; Figure S30), and [Ti(OCH₃)₄]₂ (61 and 77 ppm; Figure S31) are similar to their calculated molecular models (δ_{iso} = 61-68 ppm, 60-65 ppm, and 65-69 ppm, respectively).

For comparison with molecular models, the chemical shift of methoxy complexes of Ti, Zr, Hf, Nb and Ta as dimers were calculated. [Ti(OCH₃)₄] has a 13 C chemical shift of δ_{iso} = 67 ppm while the dimer [Ti(OCH₃)₄]₂ has 13 C chemical shifts ranging between δ_{iso} = 65-69 ppm. Thus, they are similar to the experimentally measured 13 C chemical shift of Ti(OCH₃)₄ of δ_{iso} = 65 ppm (measured in C₆D₆). The calculated 13 C chemical shift of CH₃OH coordinated to [Ti(OCH₃)₄] is δ_{iso} = 54 ppm, similar to free CH₃OH. This shows that the experimentally measured 13 C chemical shift in ssNMR of Cu/Ti@SiO₂ treated with 13 CO₂/H₂ corresponds to methoxy (OCH₃) (for the more downfield signal at 66 ppm) and CH₃OH coordinated to Ti(IV) surface sites at 49 ppm.

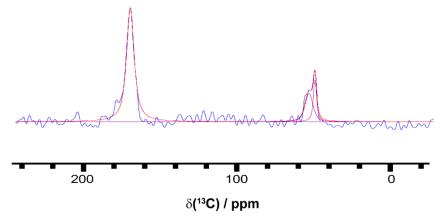
In case of $[Zr(OCH_3)_4]_2$ (δ_{iso} = 61-63 ppm), the ^{13}C chemical shift is less downfield than for $[Ti(OCH_3)_4]_x = 1$ or 2 (δ_{iso} = 65-69 ppm). The corresponding ^{13}C chemical shift in ssNMR of $Cu/Zr@SiO_2$ treated with $^{13}CO_2/H_2$ shows a major signal at 49 ppm but also shows a downfield shoulder (Figure 3 of the main text; Figure S11). Deconvolution of this feature (Figure S11) shows that the shoulder is located at 53 ppm and likely represents the Zr-OCH $_3$ intermediates (or strongly coordinated CH_3OH species), while the feature at 49 ppm is CH_3OH coordinated to Zr(IV) surface sites. The ^{13}C chemical shift for $[Hf(OCH_3)_4]_2$ (δ_{iso} = 59-61 ppm) is less downfield-shifted compared to $[Zr(OCH_3)_4]_2$ (δ_{iso} = 61-63 ppm) and more similar to that for CH_3OH (54 ppm). $Cu/Hf@SiO_2$ treated with $^{13}CO_2/H_2$ shows only one major signal at around 49 ppm, and it is therefore indistinguishable whether this signal corresponds to CH_3OH or OCH_3 coordinated (or $Si(OCH_3)_4$).

For $[Nb(OCH_3)_5]_2$ the ^{13}C chemical shift ranges between δ_{iso} = 61-68 ppm. The signal observed in ssNMR at 62 ppm for $Cu/Nb@SiO_2$ treated with $^{13}CO_2/H_2$ likely corresponds to Nb-OCH₃ intermediates along with a less downfield shifted signal at 49 ppm likely corresponding to CH_3OH coordinated to Nb surface sites. The shoulder observed between the two main signals likely corresponds to more strongly coordinated CH_3OH species on Nb(V). Similarly, for $[Ta(OCH_3)_5]_2$ the ^{13}C chemical shift is between δ_{iso} = 60-65 ppm. The signal observed in ssNMR at 59 ppm for $Cu/Ta@SiO_2$ treated with $^{13}CO_2/H_2$ likely corresponds to $Ta-OCH_3$ intermediates along, with a less downfield signal at 49 ppm likely corresponding to CH_3OH coordinated to CH_3OH species on CH_3OH species

The comparison between measured ($\delta_{iso,exp}$) and the mean of calculated ($<\delta_{iso,calc}>$) chemical shifts is shown in Figure S12.

Table S16. Calculated ¹³C chemical shifts.

Compound	¹³ C chemical shift / ppm
CH₃OH	54
[Si(OCH ₃) ₄]	53
[Ti(OCH ₃) ₄]	67
[Ti(OCH ₃) ₄ (CH ₃ OH)] [Ti(OCH ₃) ₄] ₂	54 (CH ₃ OH); 65-69 (OCH ₃) 65-69
CH ₃ O-Ti@SiO ₂	71
$[Zr(OCH_3)_4]_2$	61-63
CH ₃ O-Zr@SiO	66
$[Hf(OCH_3)_4]_2$	59-61
CH ₃ O-Hf@SiO ₂	66
[Nb(OCH ₃) ₅] ₂	61-68
CH ₃ O-Nb@SiO ₂	69
[Ta(OCH ₃) ₅] ₂	60-65
CH₃O-Ta@SiO₂	66



 $\textbf{Figure S11.} \ \ \text{Measured (blue) and deconvoluted (pink)} \ \ ^{13}\text{C ssNMR spectra for Cu/Zr@SiO}_2, \ treated \ with \ \ ^{13}\text{CO}_2/H_2.$

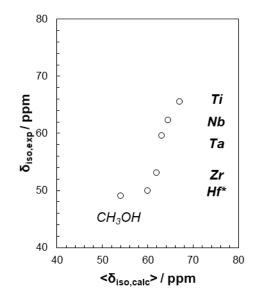


Figure S12. Mean of calculated (<δ_{iso,calc}>) and measured (δ_{iso,exp}) ¹³C chemical shifts of molecular models for Ti-OCH₃, Zr-OCH₃, Hf-OCH₃*, Nb-OCH₃, Ta-OCH₂, and CH₂OH

*For Hf-OCH₃, the chemical shift for the sole measured signal in the CH₃OH/OCH₃ region (48-66 ppm) was assumed to be the δ_{Iso,exp} value for Hf-OCH₃ and not coordinated CH₃OH. The measured signal could also arise from OCH₃ coordinated to the Si of the support or from adsorbed CH₃OH.

Natural Chemical Shielding Analysis

The CH₃OH formation rates among the catalysts with Lewis acidic Ti, Zr, Hf, Nb, and Ta surface sites in Cu/M@SiO₂ (M = Ti, Zr, Hf, Nb, Ta) show the following trend: Ti > Nb > Ta > Zr > Hf. The same trend is observed for the measured 13 C chemical shift of the methoxy intermediate coordinated to Ti, Zr, Hf, Nb, and Ta; this correlation between the CH₃OH formation rate and the 13 C chemical shift of methoxy suggests the presence of a common underlying phenomenon. The 13 C chemical shift can be directly related to the electronic structure around the nucleus, $^{[32]}$ thus allowing the development of a relationship between the obtained catalytic results and the electronic properties (i.e., Lewis acidity induced by Ti, Zr, Hf, Nb, and Ta surface sites in Cu/M@SiO₂).

To model the methoxy ligand coordinated to the Lewis acidic surface sites (Ti, Zr, Hf, Nb, and Ta), cluster models of SiO_2 with a metal site (Ti, Zr, Hf, Nb, and Ta) coordinated by three (Ti, Zr, Hf) or four (Nb, Ta) siloxy ligands, and one methoxy ligand were constructed (Figure S13). The calculated 13 C chemical shifts (δ_{iso}) are in agreement and follow similar trends as with the experimentally measured 13 C chemical shift of methoxy in the CP-MAS NMR, albeit with chemical shift values that are similar among the three least deshielded metals (Ta, Zr and Hf) (Figure S13). The 13 C chemical shift (δ_{iso}) is further deconvoluted to obtain a clearer understanding of its relationship with the electronic structure surrounding the methoxy carbon.

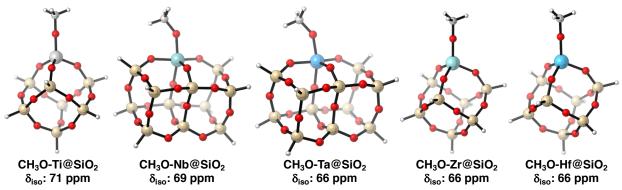


Figure S13. Cluster models of M@SiO₂ (M = Ti, Zr, Hf, Nb, Ta) with a methoxy ligand coordinated to the metal (M) site and the isotropic chemical shift δ_{lso} of the carbon of methoxy obtained by scalar relativistic DFT calculations.

The isotropic ¹³C chemical shift (δ_{iso}) is the average of the three principal components of the chemical shift tensor: $\delta_{iso} = (\delta_{11} + \delta_{22} + \delta_{33})/3$ (Eq. 2

where individual components are denoted by δ_{ii} (i = 1, 2, or 3). The values of δ_{ii} can be obtained from the chemical shielding tensor principal components σ_{ii} by DFT calculations, with $\delta_{ii} \approx \sigma_{iso,ref}$ - σ_{ii} . The chemical shielding tensor (σ_{iso}) can be further split into diamagnetic (σ_{dia}) and paramagnetic (σ_{para}) contributions: $\sigma_{iso} = \sigma_{dia} + \sigma_{para}$. The former arises from the core orbitals and is rather insensitive to changes in the electronic environment around the nucleus. In contrast, the latter is sensitive to the frontier orbitals (i.e., the orbitals close to the HOMO-LUMO gap) and therefore the changes in the electronic environment around the nucleus. More precisely, the paramagnetic contribution arises from magnetically induced admixture of the ground state with the excited states via the angular momentum operator (\hat{L}_i) that acts in the direction of σ_{ii} . The effect of the angular momentum operator can be understood conceptually by the orbital-rotation model: deshielding (the decrease in chemical shielding value/increase in chemical shift value) of the chemical shielding tensor principal components (σ_{ii}) occurs when an occupied orbital can be superimposed with a virtual orbital via rotation of 90° along the direction of the angular momentum operator (Figure S14a).^[33] The deshielding is greater if the difference in energy between the occupied and virtual orbital is small, hence its effect is more pronounced for frontier orbitals.^[33] The link between the chemical shielding and frontier orbitals can be made by natural chemical shielding (NCS) analysis that allows deconvolution of the chemical shielding to contributions of the natural localized molecular orbitals (NLMO).^[27]

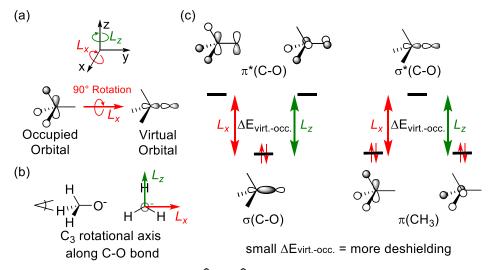


Figure S14. (a) Orientation of the angular momentum operators \widehat{L}_{χ} and \widehat{L}_{Z} . (b) Visualization of the C₃ rotational axis with respect to the angular momentum operators \widehat{L}_{χ} and \widehat{L}_{Z} . (c) Molecular orbitals mainly responsible for deshielding in case of CH₃O⁻. (Left) Coupling between the occupied σ (C-O) and the virtual π *(C-O) orbitals via \widehat{L}_{χ} and \widehat{L}_{Z} . (Right) Coupling between the occupied π (CH₃) and the virtual σ *(C-O) orbitals via \widehat{L}_{χ} and \widehat{L}_{Z} .

The principal components of the chemical shielding tensor have a similar orientation for all the cluster models (Figure S15). σ_{11} and σ_{22} are both directed perpendicular to the C-O bond (Figure S14a and S15) with σ_{11} being directed along the x-direction and σ_{22} along the z-direction, and corresponds to the angular momentum operator \hat{L}_x and \hat{L}_z , respectively. σ_{33} is oriented along the C-O bond in the y-direction corresponding to \hat{L}_y (Figure S15). The difference in the isotropic chemical shielding among the cluster models is mainly a result of the paramagnetic contribution of the two principal components σ_{11} and σ_{22} (Figure S17-19).

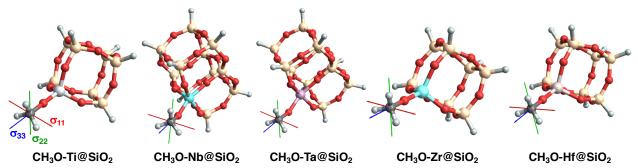


Figure S15. Chemical shielding tensor orientation (σ_{11} : red, σ_{22} : green, σ_{33} : blue) for the M-OCH₃ cluster carbon.

Further analysis of the contribution of each individual NLMO of $\sigma_{11\text{-para}}$ (Figure S20) and $\sigma_{22\text{-para}}$ (Figure S21) shows that the main contribution originates from the $\sigma(\text{C-O})$ orbital and remains constant among the different metals. The second largest contribution arises from the C-H bonding NLMO and corresponds to the $\pi(\text{CH}_3)$ molecular orbitals. These values become less negative in the order: Ti, Nb, Ta, Zr, and Hf, similar to trends observed for CH₃OH formation rates. It should be noted that \hat{L}_x and \hat{L}_z , are complementary to each other and couple similar types of orbitals due to the C₃ rotational axis of the methoxy ligand (Figure S14b and S16). The large contribution from the $\sigma(\text{C-O})$ orbital is the result of the coupling with the $\pi^*(\text{C-O})$ orbital, which is perpendicular both to the direction of the \hat{L}_x and \hat{L}_z operators (Figure S14c, S22, and S23). The $\pi(\text{CH}_3)$ orbitals are coupled with the $\sigma^*(\text{C-O})$ orbital through \hat{L}_x and \hat{L}_z (Figure S14c). The decreased (less negative) contribution correlates with the binding strength of the OCH₃ on M (in the order: M = Ti, Nb, Ta, Zr, Hf). A stronger M-O bond weakens the C-O bond and leads to a decrease in the $\sigma^*(\text{C-O})$ orbital energy and thus a smaller energy gap and more deshielding (Figure S22 and S23). NCS analysis therefore reveals that the change in ¹³C chemical shift of a methoxy ligand coordinated to different metal sites is directly related to the binding strength (i.e., the Lewis acid strength of the metal) and affects the ¹³C chemical shift, leading to a more downfield chemical shift with increasing Lewis acid strength.

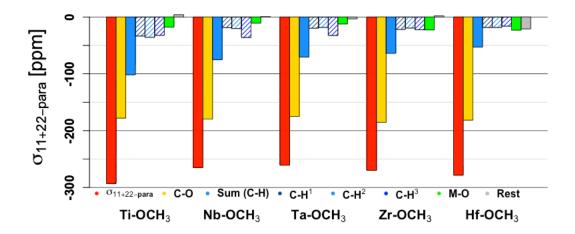


Figure S16. Sum of the paramagnetic chemical shielding contribution of $\sigma_{11\text{-para}}$ and $\sigma_{22\text{-para}}$ for the methoxy carbon of the cluster models split into the individual NLMO contributions.

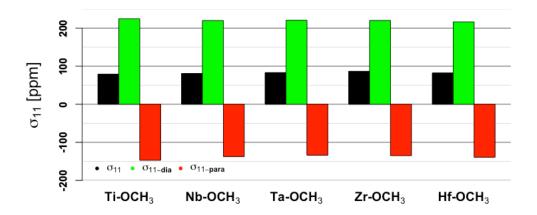


Figure S17. Chemical shielding contribution of σ_{11} for the methoxy carbon of the cluster models split in the diamagnetic ($\sigma_{11\text{-dia}}$) and paramagnetic (σ_{11} -para) contribution.

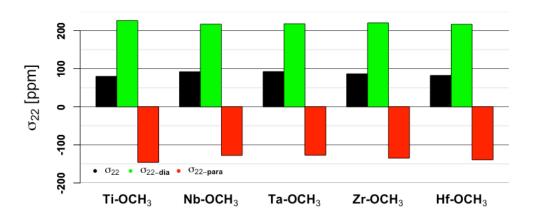


Figure S18. Chemical shielding contribution of σ_{22} for the methoxy carbon of the cluster models split in the diamagnetic ($\sigma_{22\text{-dia}}$) and paramagnetic ($\sigma_{22\text{-para}}$) contribution

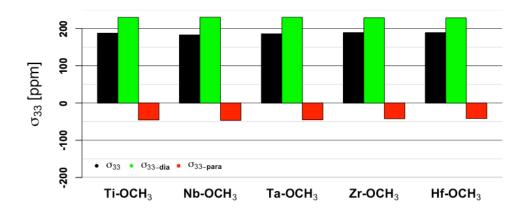


Figure S19. Chemical shielding contribution of σ_{33} for the methoxy carbon of the cluster models split in the diamagnetic ($\sigma_{33\text{-dia}}$) and paramagnetic ($\sigma_{33\text{-para}}$) contribution

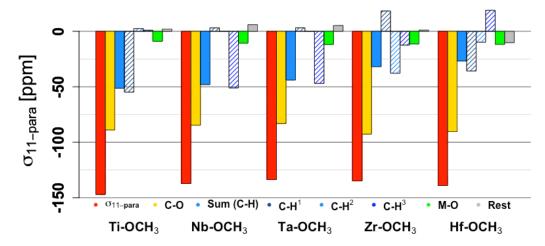


Figure S20. Paramagnetic chemical shielding contribution of $\sigma_{11\text{-para}}$ for the methoxy carbon of the cluster models split into the individual NLMO contributions.

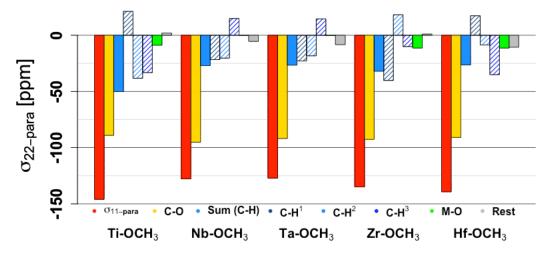


Figure S21. Paramagnetic chemical shielding contribution of $\sigma_{22\text{-para}}$ for the methoxy carbon of the cluster models split into the individual NLMO contributions.

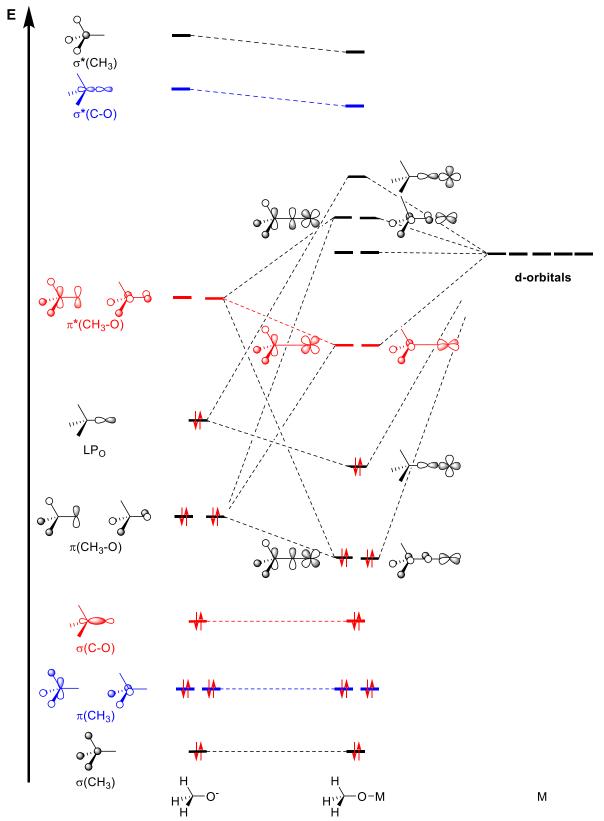


Figure S22. MO scheme of the σ and π interaction between the methoxy ligand and d^0 metal. The orbitals responsible for the deshielding of the σ (C-O) orbital (red) and π (CH₃) orbitals (blue) with virtual orbitals in σ_{11} and σ_{22} are highlighted.

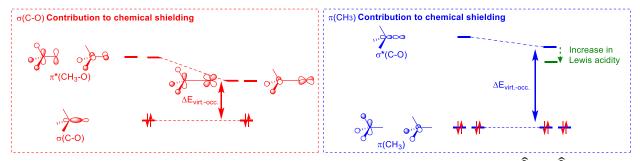


Figure S23. (Left) The σ (C-O) contribution through coupling with the π *(CH₃-O) orbitals by the angular momentum operator $\widehat{L}_{\mathcal{X}}$ and \widehat{L}_{Z} leading to chemical shielding in σ ₁₁ and σ ₂₂. (Right) The π (CH₃) contribution through coupling with the σ *(C-O) orbitals by the angular momentum operator $\widehat{L}_{\mathcal{X}}$ and \widehat{L}_{Z} leading to chemical shielding in σ ₁₁ and σ ₂₂. The decrease in energy of the σ *(C-O) orbital by increased Lewis acidity is highlighted in green.

Table S17. Isotropic and principal components of the chemical shift and shielding of the cluster models.

	$\delta_{iso}^{(a)}$ [ppm]	თ _{iso} [ppm]	₀₁₁ [ppm]	თ₂₂ [ppm]	σ₃₃ [ppm]	$\sigma_{11}^{(b)}$ [ppm]	$\sigma_{22}^{(b)}$ [ppm]	თ ₃₃ ^(b) [ppm]
Ti-OCH ₃	71	116	79	80	188	79	80	188
Nb-OCH ₃	69	118	79	91	185	81	92	183
Ta-OCH ₃	66	121	82	92	188	83	92	186
Zr-OCH ₃	66	121	86	86	190	87	87	189
Hf-OCH ₃	66	121	86	86	191	82	82	189

⁽a) Referenced to tetramethylsilane.

Table S18. Diamagnetic and paramagnetic chemical shielding contribution of the cluster models.

Ti-OCH ₃	σ₁₁₋dia [ppm]	σ _{11-para} [ppm]	σ _{22-dia} [ppm]	σ _{22-para} [ppm]	σ _{33-dia} [ppm]	σ _{33-para} [ppm]
Nb-OCH ₃	225	-147	226	-146	230	-45
Ta-OCH₃	220	-137	217	-128	231	-46
Zr-OCH ₃	221	-134	218	-127	230	-45
Hf-OCH ₃	220	-135	220	-135	229	-42
Ti-OCH ₃	216	-139	216	-139	229	-42

Table S19. NLMO contribution to the paramagnetic chemical shielding ($\sigma_{11\text{-para}}$) of the cluster models.

	σ _{11-para} [ppm]	C-O [ppm]	Sum (C-H) [ppm]	C-H ¹ [pmm]	C-H² [ppm]	C-H ³ [ppm]	M-O [ppm]	Rest [ppm]
Ti-OCH ₃	-147	-89	-51	-55	3	1	-9	2
Nb-OCH ₃	-137	-85	-48	3	0	-51	-11	6
Ta-OCH₃	-134	-83	-44	3	0	-47	-12	5
Zr-OCH ₃	-135	-93	-32	18	-38	-12	-11	1
Hf-OCH ₃	-139	-90	-27	-36	-10	19	-12	-10

Table S20. NLMO contribution to the paramagnetic chemical shielding ($\sigma_{22\text{-para}}$) of the cluster models.

		σ _{22-para} [ppm]	C-O [ppm]	Sum (C-H) [ppm]	C-H ¹ [ppm]	(C-H² [ppm]	C-H ³ [ppm]	M-O [ppm]	Rest [ppm]
•	Ti-OCH₃	-146	-89	-50	21		-38	-33	-9	2
١	Nb-OCH₃	-128	-95	-27	-22		-20	15	0	-5
7	Га-ОСН₃	-127	-92	-27	-23		-18	15	0	-8
2	Zr-OCH₃	-135	-93	-32	-40		18	-10	-11	1
ŀ	Hf-OCH₃	-139	-91	-26	17		-9	-35	-11	-11
XYZ	Coordinates	for Molecular N	l odels			Н	1.3553636	-1.7779175	1.1398312	
TMS	3					Н	2.4034086	-0.6780315	0.2458472	
Si	0.0001166	0.0007625	-0.0002818			Н	1.4933046	-1.9205905	-0.6117198	
С	-0.0315924	1.2116905	1.4514412			С	0.2120466	0.9583435	-1.6165758	
Н	0.8949666	1.7911615	1.5138922			Н	-0.6073534	1.6663175	-1.7764958	
Н	-0.1543884	0.6917915	2.4068812			Н	0.2357366	0.2878775	-2.4814498	
Н	-0.8562544	1.9258295	1.3617112			Н	1.1437846	1.5329145	-1.6255568	
С	-1.6235054	-0.9671875	-0.0477218							
Н	-1.7818114	-1.5332975	0.8756672			CH ₃ C	DΗ			
Н	-1.6422394	-1.6838115	-0.8750288			С	0.2932617	-0.2062687	-0.0174933	
Н	-2.4848144	-0.3034285	-0.1733098			0	-0.4993843	0.9093303	0.3545867	
С	1.4432316	-1.2024235	0.2128682			H	-0.1055683	-1.1538507	0.3698547	

⁽b) Sum from NCS analysis

Н	1.2803637		0.4185547		OCH ₃) ₄] ₂		
H H	0.4106117 -1.3792843		1.1062583 0.0192443	Ti O	0.8976559 0.5911229	0.8565690 2.4257720	-1.1014163 -1.9579473
				0	0.8223369	-0.8613280	0.1738357
[Si(C	OCH ₃) ₄] 0.00012	0.00027 -0.00	1072	0	1.0994089 2.4420449	-0.2394160 1.3450490	-2.5309353 -0.3015483
0	1.13835	-0.85474 -0.8		C	3.2261319	1.2707880	0.8590787
0	0.79423		1392	Н	4.2347669	0.9240860	0.6082537
0	-0.83478		1420	Н	2.7847159	0.5816130	1.5876237
O C	-1.09747 -2.16311	0.76959 -0.94 0.15223 -1.65		H C	3.3076779 1.3445189	2.2600530 -0.1040150	1.3225787 -3.9086633
H	-2.79454	0.94293 -2.06		H	2.4101019	-0.2442610	-4.1234893
Н	-2.76186	-0.48162 -0.99		Н	1.0406489	0.8873930	-4.2640283
H C	-1.78168 0.19633	-0.45519 -2.48 2.27300 1.49		H C	0.7781299 1.7709759	-0.8642770 -1.9159670	-4.4570023 0.0657887
H	0.99780		1206	H	1.6217659	-2.4540230	-0.8738823
Н	-0.39028	1.94972 2.36	381	Н	1.6587259	-2.5994960	0.9086127
H C	-0.45361 -0.26927		1595 9914	H C	2.7818669 1.0704129	-1.4993790 3.7440130	0.0687597 -1.9884603
Н	0.27447		5790	Н	1.2357969	4.0575810	-3.0249253
Н	0.41282	-2.73224 1.10	0546	Н	2.0154679	3.8213710	-1.4381213
Н	-1.08663		3517	Н	0.3413659	4.4271840	-1.5367483
C H	2.23568 2.80754	-0.30784 -1.53 0.38722 -0.91		H H	-1.6217271 -2.4105241	2.4537990 0.2452360	0.8750187 4.1230297
H	1.89238	0.21650 -2.43		H	-1.2321341	-4.0603760	3.0223017
Н	2.88150	-1.13365 -1.83		Н	-4.2351481	-0.9246000	-0.6087903
T:/C	2011.1			Ti	-0.8976471	-0.8559750	1.1018607
Ti	0.0012607	0.0002940	0.0004729	C C	-1.0703321 -3.2260181	-3.7441130 -1.2708630	1.9860937 -0.8582253
Ö	0.5458897		0.9857619	Č	-1.3449171	0.1033860	3.9094207
0	-0.4642153		1.0504139	C	-1.7707911	1.9165830	-0.0651583
O C	-1.4290333 -2.1986303		-0.9671431 -1.5766801	0	-0.8225371 -0.5910841	0.8616440 -2.4258120	-0.1737213 1.9574007
Н	-2.1980303		1.0418781	Ö	-1.0979011	0.2393110	2.5320527
Н	-1.8812203		2.6150661	0	-2.4431271	-1.3433520	0.3033197
Н	-3.2538193		1.5747591	Н	-1.6579191	2.6007860	-0.9073543
C H	-1.4455603 -2.4107803		1.7231239 1.2081929	H H	-1.0430831 -0.3429101	-0.8887730 -4.4261690	4.2643997 1.5300677
H	-1.1485463		1.7672059	H	-3.3064461	-2.2606360	-1.3208553
Н	-1.5668723	1.7538520	2.7475519	Н	-2.7818461	1.5003900	-0.0689303
С	1.5723827		1.5751859	Н	-2.0172721	-3.8200290	1.4387997
H H	1.2798867 1.7647417		1.6214159 2.5949049	H H	-2.7843611 -0.7779121	-0.5820850 0.8623390	-1.5870133 4.4589197
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	₃O-Ti@SiO₂]	or the silied cids	ici illoucis	ŏ	-0.8668059	2.6692517	0.1648079
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O	-1.6050589	-0.0634443	-1.5359170	Ö	-0.8453289	-1.1981243	-2.3978241
Si	0.9423571	2.6118037	2.1982940	Ō	-0.8617259	1.4792137	-2.2338951
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Si	-1.8356609	1.5142477	-1.1644490	Н	0.3130741	-0.2545753	4.1200199
0	1.1601791	-1.0525483	-1.4808050	Н	-5.0535159	-0.0296133	-0.0160271
Ti	-0.5712779	-1.5430783	-1.3223430	H O	-2.2358959	0.2334337	-3.9636711
O H	-2.2683879 -3.0990529	1.6361277 1.7516187	0.4161740 2.8062330	C	4.1564241 5.5598411	0.0224307 0.0283927	0.0172219 0.0297999
Si	2.2218061	-0.2819363	2.1482940	Н	5.9289611	-0.0514943	1.0568339
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Ö	-0.4261619	2.3249017	-1.4076270	H	5.9465181	-0.8158222	-0.5491561
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Н	3.6410201	-0.4753843	-1.7749810	[CH:	3O-Nb@SiO2]		
Si	1.0964421	2.7291507	-0.9547430	Ō	-3.4577038	0.4071864	2.1186777
Н	-2.8854319	2.1028467	-1.9970590	Si	-0.7353868	-2.2193856	2.8938587
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С	-1.3855359 -1.6647709	-3.8944093 -4.7810973	-3.2761420 -2.6938050	Si O	-0.6314858 1.0175472	-1.6436926 0.8448324	-1.4734263 1.5481997
H H	-0.5481559	-4.1491803	-3.9343990	Si	-4.1347148	-0.2259026	0.7597747
H	-2.2463159	-3.5833903	-3.8808990	0	0.4362592	-1.2410226	-0.3062253
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O	-3.0325286	0.3464980	1.5047240	Si	-0.2368488	1.8606654	1.1584037
Si	0.0228054	-1.9664740	1.8512010	0	0.7738692	-2.0275306	2.2827177
Si	-1.7672146	1.8798520	-1.7691170	Nb	1.7517452	-0.8254746	1.0933527
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0	-3.0360216	1.1186060	-1.0555040	Si	-2.5079228	-3.0170346	0.3780487
0	-0.8577716	0.7709170	-2.5658530	Н	-3.2774948	-4.2518006	0.2635257
Si	0.0185044	-0.6202560	-2.6268510	0	-1.1957008	1.9032154	2.4826517
O Si	1.3889244 -3.5940776	1.7206760 -0.0054060	0.5159130 0.0029390	O H	-1.3336248 -0.7123128	-0.7115186 -2.9976576	3.2061427 4.1318937
0	1.3970744	-0.4156690	-1.7597340	Si	-2.1715328	0.6925914	3.1011477
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ŏ	-0.8625446	-2.6092260	0.6224290	H	-2.6324998	1.1422504	4.4106057
Si	0.0177564	2.5845850	0.7766920	0	3.3403762	-1.0762656	2.1093647
0	1.3968044	-1.3171180	1.2305670	С	4.7428262	-1.1703106	2.0313217
Zr	2.2203164	-0.0018760	-0.0037140	Н	5.0570892	-2.1981246	2.2316027
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Si	-1.7662516	-2.4766940	-0.7404780	0	2.9830942	-0.3381856	-0.3763203
H O	-2.2488706 -0.8612796	-3.7961330 1.8411350	-1.1385260 1.9522720	O Si	0.1825042 3.0480452	-1.6592106 0.7086084	-2.9017823 -1.6303993
Ö	-0.8499886	-0.7711970	2.5699680	Si	0.8847352	-0.3694316	-3.6872973
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Si	-1.7620036	0.5915250	2.5168630	ŏ	2.3170182	2.1115914	-1.1490463
Н	0.3191834	3.9553420	1.1888060	H	4.4331032	0.9966624	-2.0070693
Н	0.3082724	-0.9514400	-4.0218470	0	-0.0813518	0.9680974	-3.6735643
Н	-5.0539876	-0.0014600	0.0060860	Н	1.1602162	-0.7821136	-5.0598523
Н	-2.2514936	0.9064810	3.8561950	0	-2.1747188	2.0127704	-2.3719283
0	4.1620964	0.0042960	-0.0033230	0	0.4607462	3.1913884	0.5121267
С	5.5666694	0.0111170	-0.0060480	Si	1.2433212	3.3341764	-0.9608973
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Н	5.9392264	0.9943800	0.2968630	H H	1.9023742 -1.1387168	4.6363364 3.1816844	-0.9691013 -4.3863853
ſСН	3O-Hf@SiO2]			- 11	-1.1307 100	J. 10 100 44	-4.5005055
0	-3.0399139	0.0799347	-1.5596731	[CH	₃O-Ta@SiO₂]		
Si	0.0131371	2.4160747	-1.2023231	O	-3.4573459	0.4138564	2.1182657
Si	-1.7537889	-2.3231952	1.1433669	Si	-0.7435219	-2.2183666	2.8997277
0	-0.8783649	1.1874837	2.4034769	Si	-2.2693949	0.9057924	-1.1735783
0	-3.0265879	-1.3982903	0.6710429	0	-1.4782839	-2.9564496	-0.9252893
0	-0.8588349	-1.4851973	2.2337509	0	-3.7455589	0.7835204	-0.4965183
Si	0.0118461	-0.1661683	2.6915239	0	-1.8050019	-0.5417376	-1.7291093
O Si	1.4087271	-1.4928933	-0.9946831	Si O	-0.6356079	-1.6576766	-1.4819833
31	-3.5936339	-0.0198953	-0.0169891	J	1.0265821	0.8331004	1.5388817

Si -4.1344659 -0.2240096 0.7619997 O 0.4446121 -1.2750126 -0.3192353 O -3.5216119 -1.7139056 0.4381557 O -1.6719329 -2.9966466 1.7886457 Si -0.2244069 1.8502274 1.1542417 O 0.7621051 -2.0170056 2.2907577 Ta 1.7594361 -0.8464236 1.0775487 O -1.1516799 1.2312694 -0.0255223 Si -2.5128989 -3.0185686 0.3796977 H -3.2853869 -4.2519436 0.2697357 O -1.1847749 1.8985504 2.4776667 O -1.3442769 -0.7119196 3.2120837 H -0.7265409 -3.0001686 4.1354197 Si -2.1701869 0.6980124 3.1001857 H -5.5848879 -0.2840856 0.9100407 H -2.6293929 1.1575754 4.4068347 O 3.3478361 -1.1203366 2.1011577 C 4.7548201 -1.1637486 2.0689817	нноояяоононоояяонн	5.1487841 5.1149491 2.9801691 0.1761161 3.0504291 0.8786261 2.2639151 2.3316941 4.4341611 -0.0895319 1.1572671 -2.1765509 0.4673751 1.2407471 -0.8024389 0.1761341 1.8824531 -1.1508829	-0.5020926 -0.8354166 -0.3054566 -1.6650966 0.7260684 -0.3715036 0.0738914 2.1377924 1.0019544 -0.7828686 2.0073704 3.1820824 3.3425784 2.3727034 3.2334254 4.6532794 3.1760284	2.8444837 1.0889717 -0.3836873 -2.9121353 -1.6478403 -3.6928573 -2.9359753 -1.1757073 -2.0370613 -3.6788123 -5.0651793 -2.3693923 0.5033547 -0.9720903 -3.2230383 -2.2198763 -0.9765873 -4.3903103
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Section S10. ATR-IR spectra and solution 1 H, 13 C, and 1 H- 13 C HSQC NMR spectra of [Nb(OCH₃)₅]₂ and [Ta(OCH₃)₅]₂. CP-MAS 13 C-NMR spectrum for [Ti(OCH₃)₄]₂.

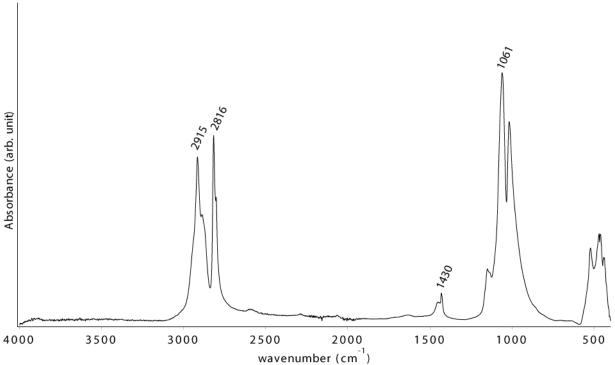


Figure S24. ATR-IR spectrum of [Nb(OCH₃)₅]₂ (298 K, under inert atmosphere).

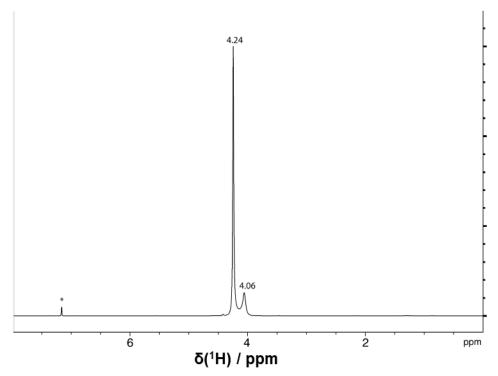


Figure S25. Solution ¹H NMR spectrum of [Nb(OCH₃)₅]₂ (C₆D₆).

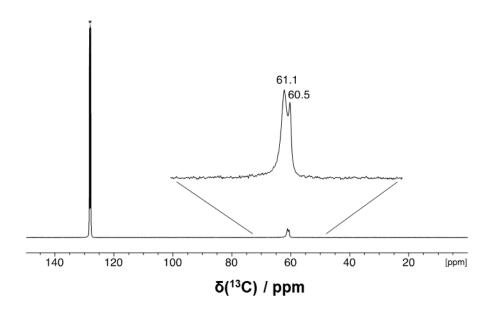


Figure S26. Solution ^{13}C NMR spectrum of [Nb(OCH $_3$) $_5$] $_2$ (C $_6$ D $_6$).

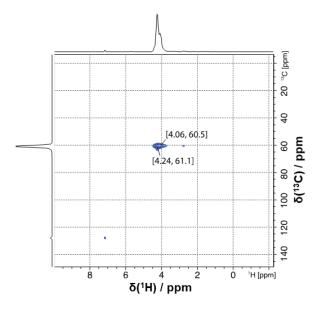
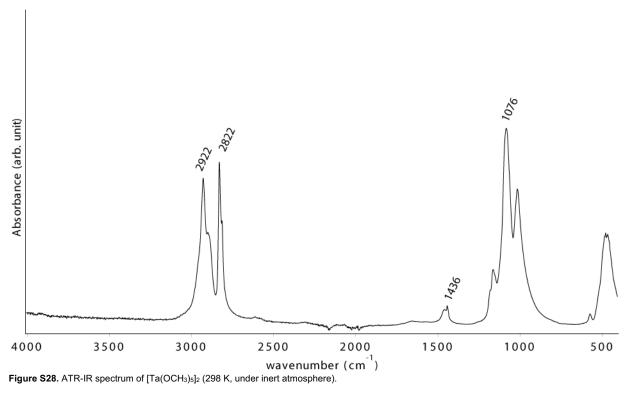


Figure S27. Solution $^1H-^{13}C$ HSQC NMR spectrum of [Nb(OCH₃)₅]₂ (C₆D₆).



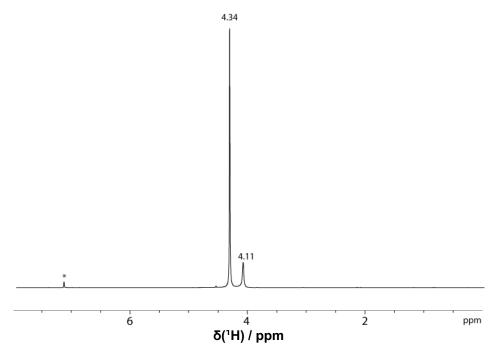


Figure S29. Solution 1H NMR spectrum of [Ta (OCH₃)₅]₂ (C₆D₆).

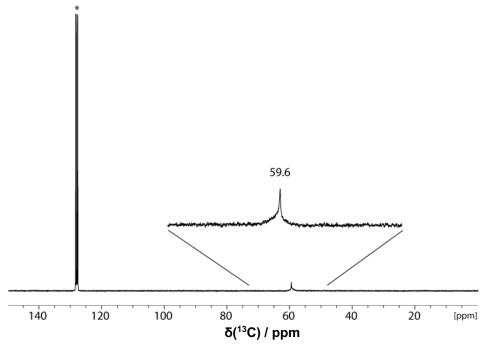
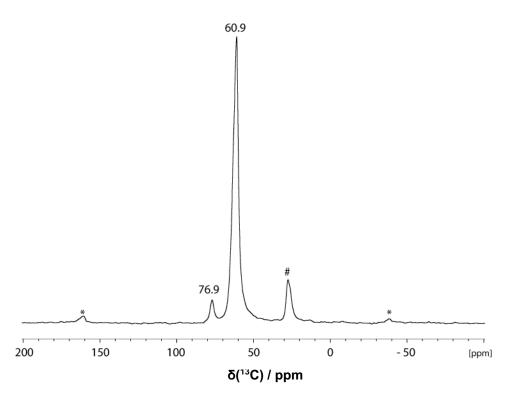


Figure S30. Solution ^{13}C NMR spectrum of $[Ta(OCH_3)_5]_2$ (C_6D_6) .



 $\label{eq:Figure S31.} \textbf{Figure S31.} \textbf{ Solid state CP-MAS} \ ^{13}\textbf{C NMR spectrum of } [\text{Ti}(\text{OCH}_3)_4]_2 \ (\text{spinning rate = 10 kHz}). \\ ^* \textbf{ denotes spinning sidebands}; \# \textbf{ denotes an unknown impurity in the as-purchased compound.} \\$

Section S11. CH_3OH formation rates as a function of pyridine adsorption enthalpies for $Cu/M@SiO_2$

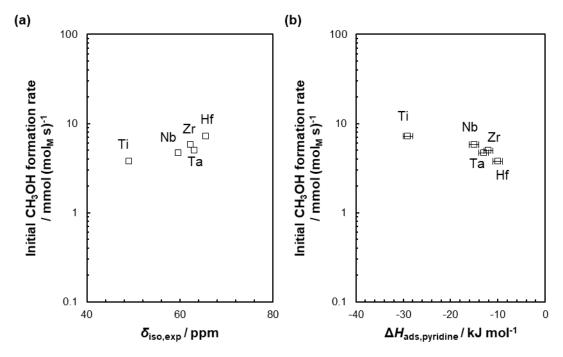


Figure S32. Initial CH₃OH formation turnover rate, normalized by moles of M, for Cu/M@SiO₂ as a function of (a) the 13 C chemical shift for methoxy intermediates ($\delta_{\text{lso,exp}}$) and (b) the pyridine adsorption enthalpy for M@SiO₂. Note the logarithmic scale. Error bars in the y-value are smaller than the size of the data point.

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Author contributions

[33]

G.N. and C.C. designed research. G.N., J.M., P.S. (EXAFS fitting); E.L (DFT); and D.T.B. (pyridine enthalpies of adsorption) and D.W.F. (pyridine enthalpies of adsorption) performed research. C.C. and D.W.F. acquired funding. G.N. and C.C. wrote the original draft. All authors contributed to the analysis and interpretation of results and writing of the final manuscript.

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